



UCLA


Ryan Dumlao
University of California, Los Angeles
MS Comprehensive Exam Project
Spring 2010
Advisor: Diana Huffaker

Supporting Group: Andrew Lin, Giacomo Mariani

MODEL FOR CALCULATING THE DENSITY AND RESISTIVITY OF SURFACE STATES IN N-DOPED GAAS NANOPILLARS

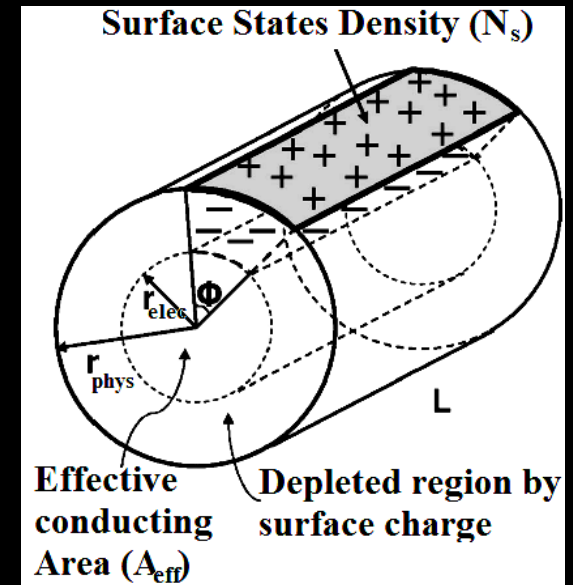


Outline

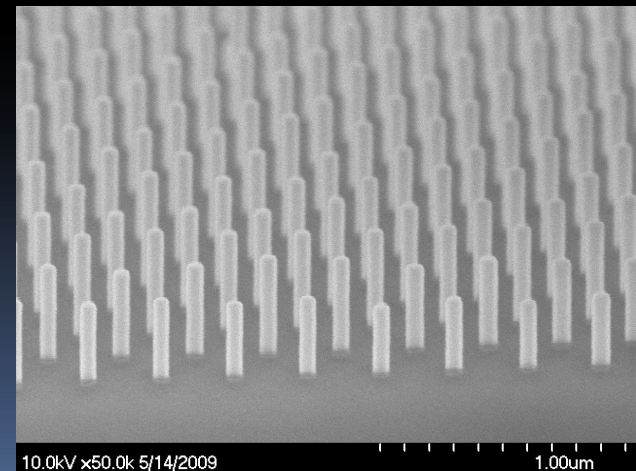
- Objective
 - Motivation
 - Background
 - Approach
 - MATLAB Model
 - Device Fabrication
 - Experimental Results
 - Summary
- 

Objective

- At the nanoscale, negligible effects on a semiconductor's electrical properties such as surface states become significant
 - Large surface area-volume ratio
- The existence of surface states dramatically changes the overall doping profile of a nanopillar
- The effective radius of a pillar decreases and changes the conductive size
- By studying resistivity and taking into account the changes due to surface states, we can estimate the actual surface state density on a nanopillar



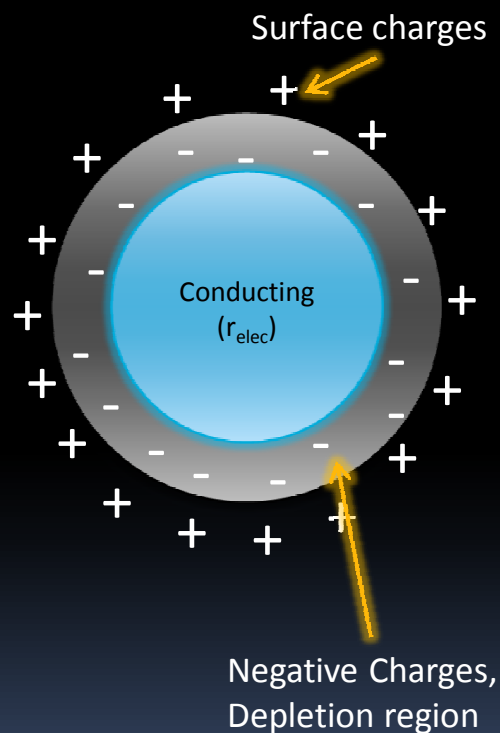
K. Seo *et al.* "Surface charge density of unpassivated and passivated metal-catalyzed silicon nanowires." *Electrochemical and Solid-State Letters*, 2006, Issue 3, Vol. 9.



Motivation

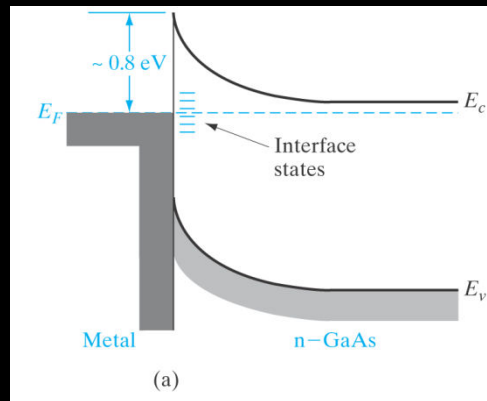
- Would like to ideally determine the surface state density N_s
- Surface state reduction techniques such as passivation need accurate measurements
- An effective, easy and simple way to determine N_s is desired
- Ability to determine density using easily taken or already existing IV measurements

Background – Surface States



- Surface states occur due to abrupt transition between solid material to outside
 - Periodicity of lattice is interrupted
 - Creates energies in the forbidden band gap of material
- Opposite charges are created within material to balance out surface charges
 - Creates a depletion region
- Transport is not available within the depletion region, effectively “shrinking” the pillar

Background – Surface States



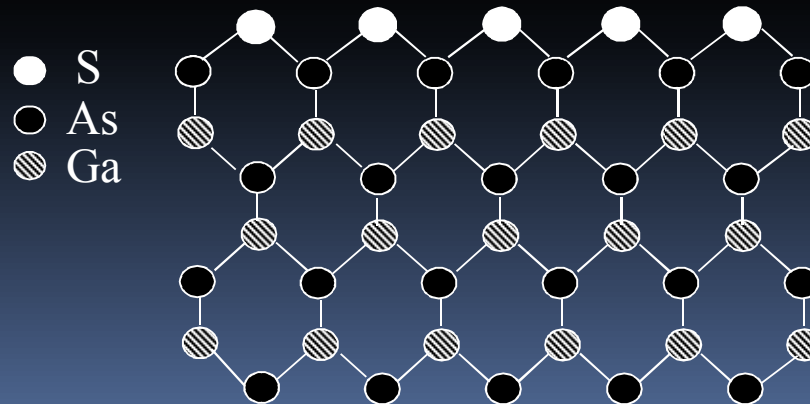
Pinned Fermi level at metal-GaAs contact

Banerjee, Sanjay and Streetman, Ben., "Solid State Electronic Devices." Upper Saddle River, NJ : Pearson Education, Inc, 2006.

- Detrimental effects of surface states
 - Energy bands within bandgap
 - Create carrier trapping centers
 - Lowers effective carrier concentration
 - Depletes the pillar - hinders carrier transport
 - Pins the Fermi level, bending conduction and valence bands
 - Creates a high Schottky barrier regardless of contact

Background - Passivation

- Passivation reduces the surface state density
- Uses replacement atoms to bind to vacancies created by the abrupt change in periodicity of lattice
- Dangling bonds are “capped” and thus charges are lessened, decreasing N_s
- Accomplished with a sulfuric solution
 - High chemical affinity between GaAs and sulfur



Approach

- Considering two radii, one physical r_{phys} and one decreased by the depletion region r_{elec} , we model the “effective” radius

- Begin with Poisson’s equation

$$\nabla^2 \psi(r) = -4\pi\rho$$

- Put into cylindrical coordinates

$$\frac{\partial^2 \psi}{\partial r^2} + \frac{1}{r} \frac{\partial \psi}{\partial r} + \frac{\partial^2 \psi}{\partial z^2} = -4\pi\rho, \quad \psi = \psi(r, z)$$

Approach

- Assuming abrupt transition in charge density ρ at radius, potential ψ within nanopillar becomes

$$\rho = \begin{cases} 0 & 0 \leq r < r_{elec} \\ q(N_D - N_A), & r_{elec} \leq r \leq r_{phys} \end{cases}$$

$$\psi(r) = \begin{cases} \psi_0 & 0 \leq r < r_{elec} \\ \psi_0 - \frac{\rho}{4\epsilon_G} (r^2 - r_{elec}^2) & r_{elec} \leq r \leq r_{phys} \end{cases}$$

- And thus the surface potential becomes

$$\psi_s = \psi_0 - \frac{\rho}{4\epsilon_G} (r_{phys}^2 - r_{elec}^2)$$

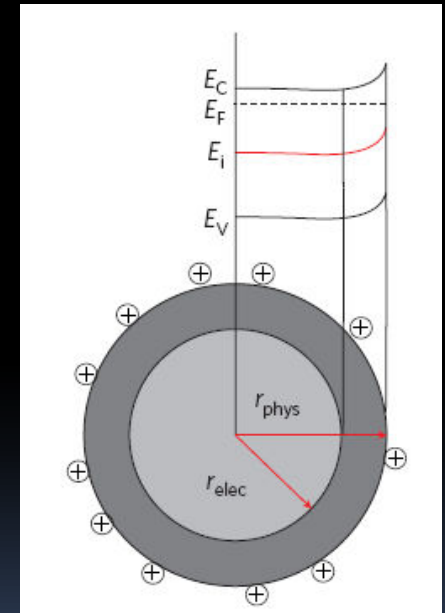
Approach

- Using charge neutrality, we find that the difference between the surface potential and the charge density below the surface must be zero

$$\pi(r_{phys}^2 - r_{elec}^2)\rho + 2\pi r_{phys}(Q_f + Q_{it}) = 0 \quad Q_{it} = -q^2 N_s \psi_s$$

- When solved for r_{elec} , we see that

$$r_{elec} = \sqrt{r_{phys}^2 + \frac{2r_{phys}Q_f - 2r_{phys}N_s\psi_0}{q(N_D - N_A)\left(1 + \frac{r_{phys}q^2}{2\epsilon_G}N_s\right)}}$$



M. T. Bjork, *et al.* "Donor deactivation in silicon nanostructures." *Nature Nanotechnology*, 2009, Vol. 4.

Approach

- Next, we must define a critical radius a_{crit} which determines if the pillar is depleted or not, when r_{elec} goes to zero

$$a_{crit} = \frac{\varepsilon_G}{q^2 N_S} \left\{ -1 + \left[1 + \frac{4q^2 N_S}{\rho \varepsilon_G} (q^2 N_S \psi_0 - Q_f) \right]^{\frac{1}{2}} \right\},$$
$$\approx \frac{2}{\rho} (q^2 N_S \psi_0 - Q_f)$$

Approach

- Depending if our physical radius is below or above the critical radius, we find the effective carrier concentration to be

$$n_{eff} = n_0 \exp(\beta\psi_0) \left\{ \frac{r_d^2}{a^2} + \frac{4\epsilon_s}{\beta\rho r_{phys}^2} \left[1 - \exp\left(\beta \frac{\rho}{4\epsilon_G} (r_{elec}^2 - r_{phys}^2) \right) \right] \right\}, r_{phys} > a_{crit}$$

$$n_{eff} = n_0 \exp(\beta\psi_s) \frac{4\epsilon_s}{\beta\rho r_{phys}^2} \left[\exp\left(\beta \frac{\rho r_{phys}^2}{4\epsilon_G} \right) - 1 \right], r_{phys} < a_{crit}$$

Approach

- Finally, we can relate this effective carrier concentration with resistivity via the mobility

$$\rho = \frac{1}{q\mu_k n_{eff}}, \quad \mu_k = \frac{\mu_0}{1 - \sqrt{\frac{n_{eff}}{10^{18}}}},$$

$$\text{where } \mu_0 = 8500 \frac{\text{cm}^2}{\text{Vs}} \text{ (n - type)}$$

- And given resistance from measured I-V curves

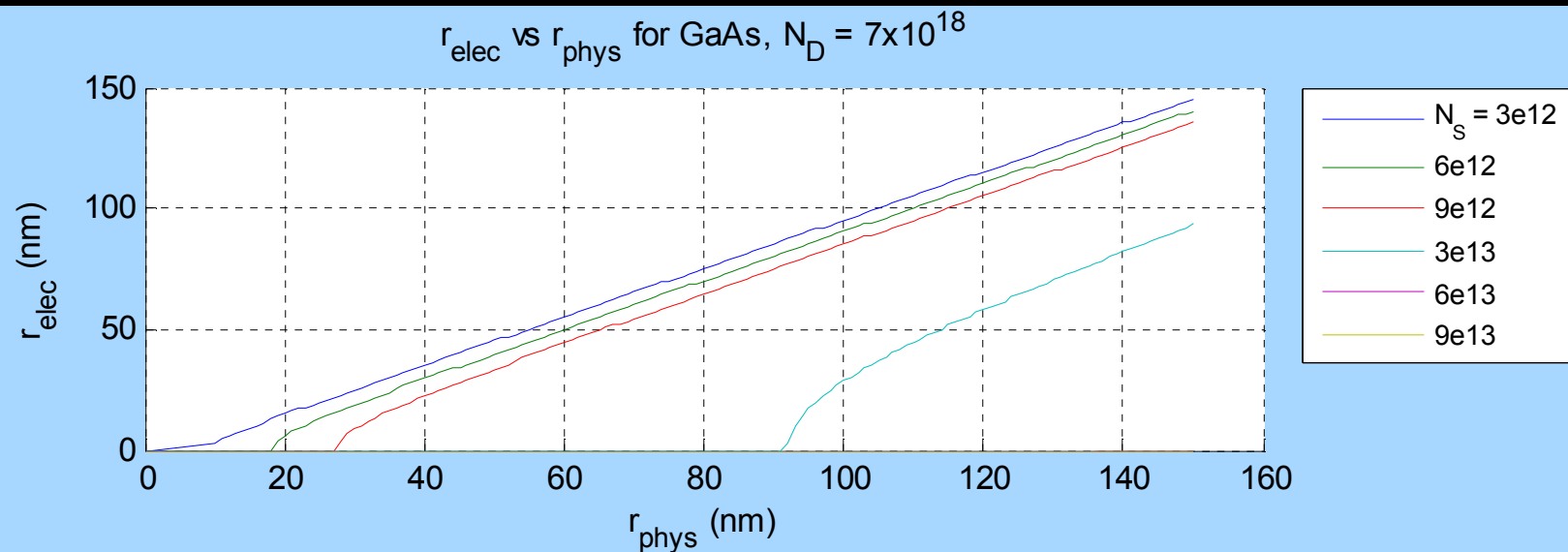
$$R = \frac{\rho L}{A_e}, \quad \text{where } A_e = \pi r_{elec}^2$$

MATLAB[®] Model

- We can now simulate the effect on the electrical radius, effective carrier concentration, and resistivity of a nanopillar due to the presence of surface states
- For simulation, we consider an n -doped GaAs nanopillar with different doping concentrations, surface state densities and radii
- For a physical device, we use an 82 nm radius , and for fixed doping, we use $7 \times 10^{18} \text{ cm}^{-3}$.
- Simulate the changes on r_{elec} , n_{eff} , and ρ due to N_s
- The simulations are then repeated, sweeping over several surface states densities (from $3 \times 10^{12} \text{ cm}^{-2}$ to $1 \times 10^{14} \text{ cm}^{-2}$)

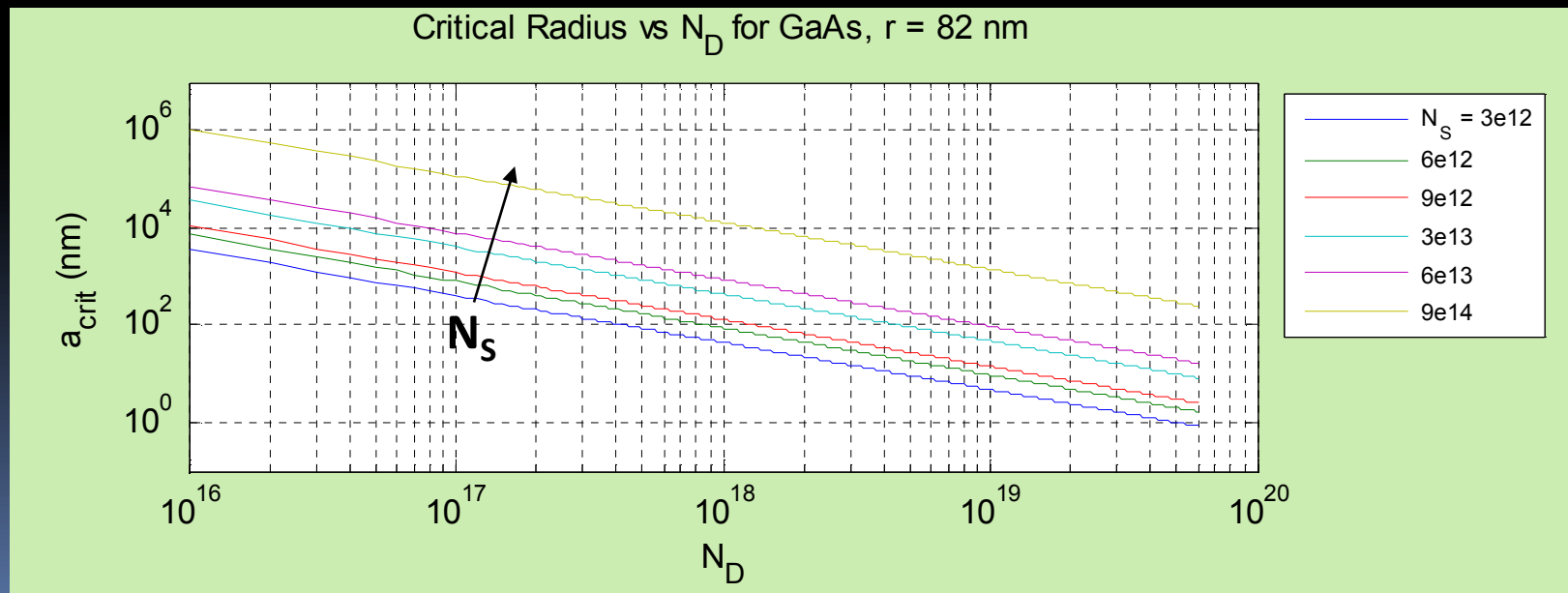
Model - Radius

- Electronic radius r_{elec} is the effective conducting radius
- For higher surface states density, the difference between physical radius and electrical radius is larger
 - Fully depleted region gets wider
- The point where r_{elec} begins is at the critical radius, determined by N_s
- Large effect of small changes in surface state density is shown



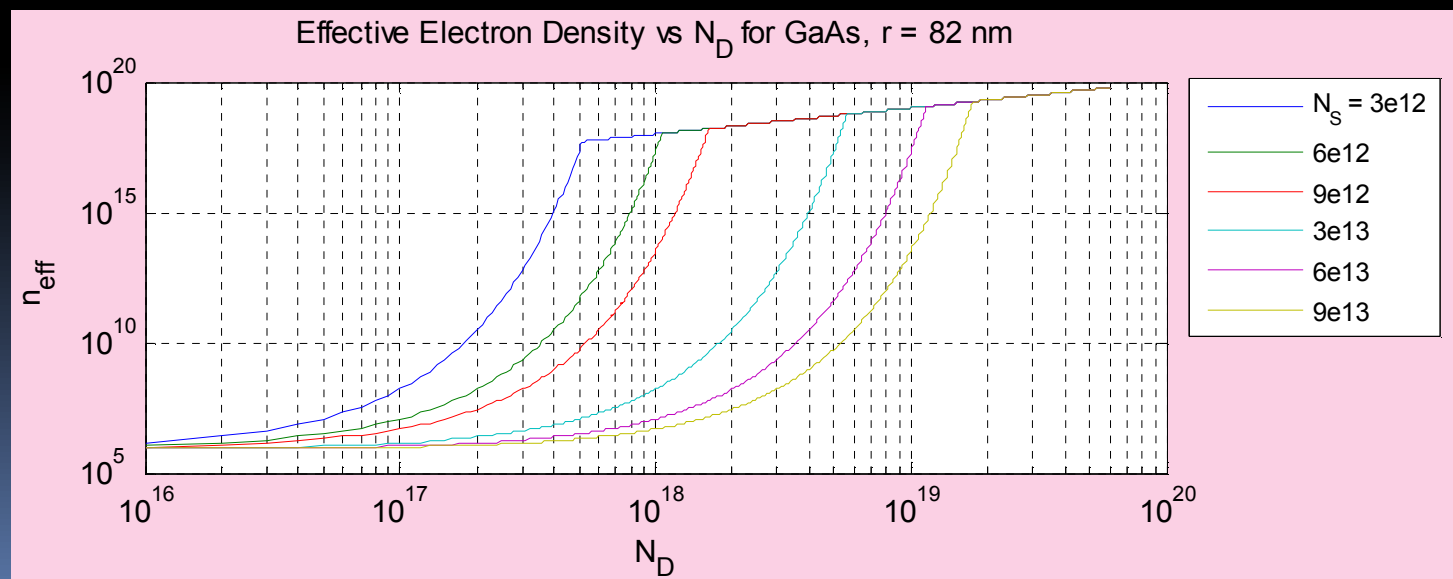
Model – Critical Radius

- Linear relationship between surface state density and depletion region
- a_{crit} decreases linearly with doping
- Value of a_{crit} corresponds to point where r_{elec} becomes non-zero



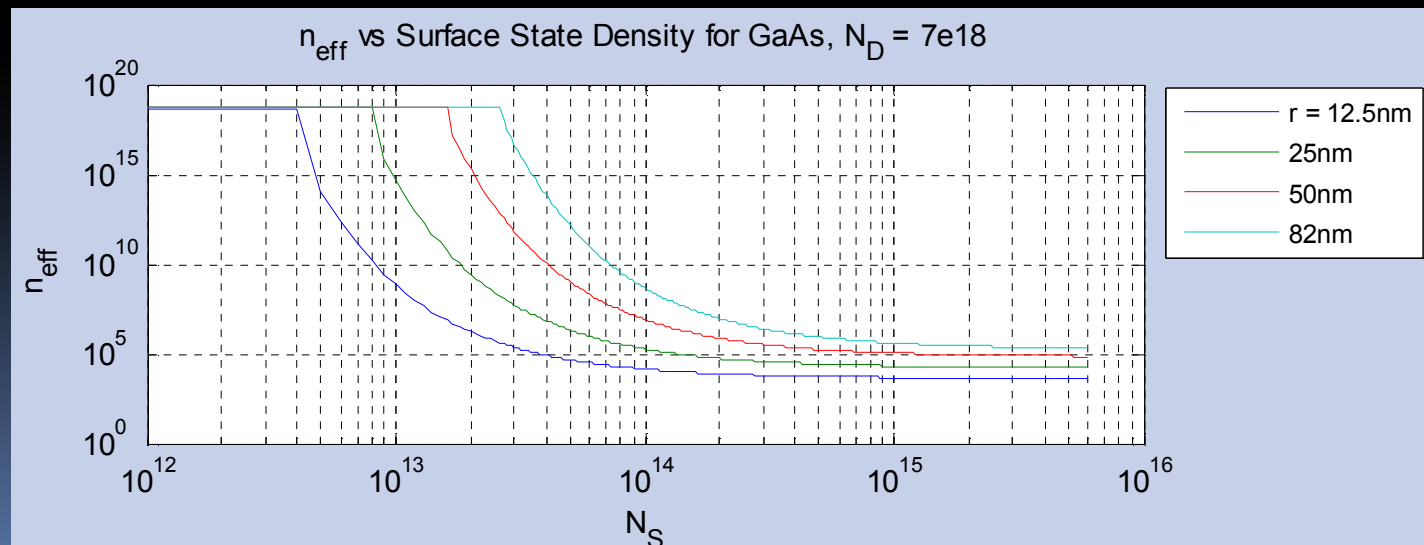
Model – Effective Carrier Concentration

- Based on the surface state density at a constant doping level, the effective carrier concentration changes
- A drastic difference (compared to actual doping) in orders of magnitude until N_S is reduced to a specific level based on N_D
 - At lowest doping, concentration reaches intrinsic value of $2.25 \times 10^6 \text{ cm}^{-3}$
 - At high doping or low N_S the pillar is almost non-depleted
- Shows a dramatic change in carrier transport ability of the nanopillar



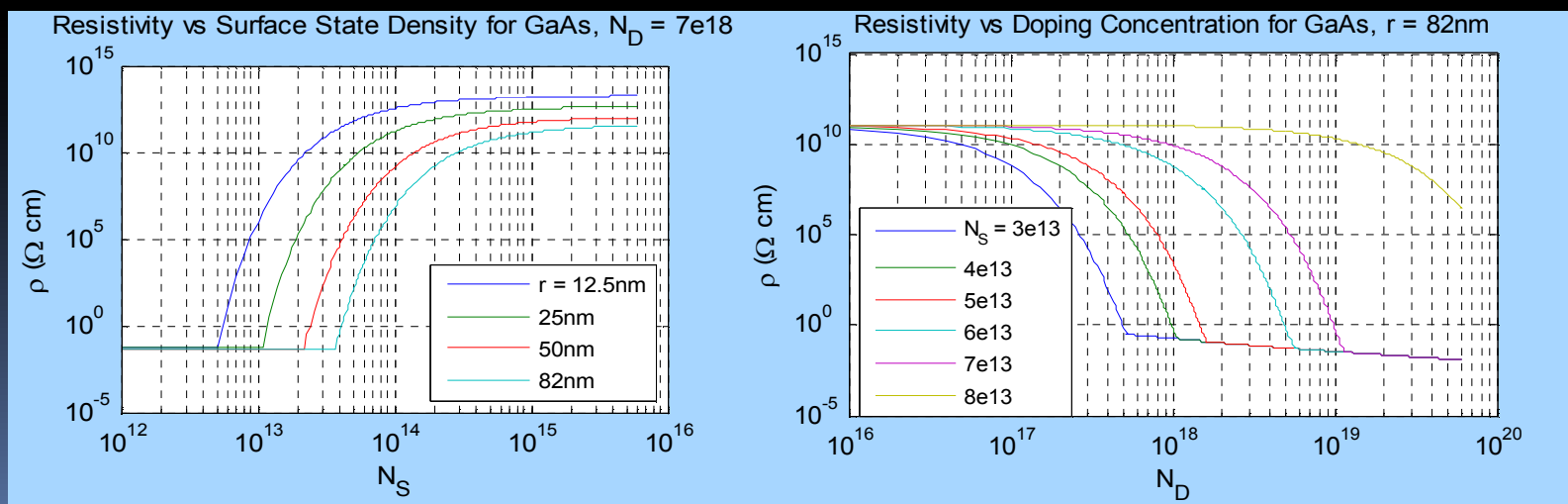
Model – Effective Carrier Concentration

- The model can also account for another case of changing surface state density with different radii (constant N_D)
- Below a certain density for each size, the surface states no longer adversely affect the carrier concentration
- We see that the smaller the nanopillar, the more affected its transport is by surface state density
 - Depletion area remains the same size for specific N_S , leaving less area for transport as pillar radius decreases



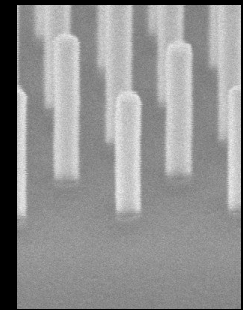
Model - Resistivity

- Relating n_{eff} to resistivity, we can model the behavior of N_S or N_D
- As surface state density decreases, so does resistivity
 - Depletion region decreases, more carrier transport
- Can match these values to resistivity gained from I-V measurements on real pillars

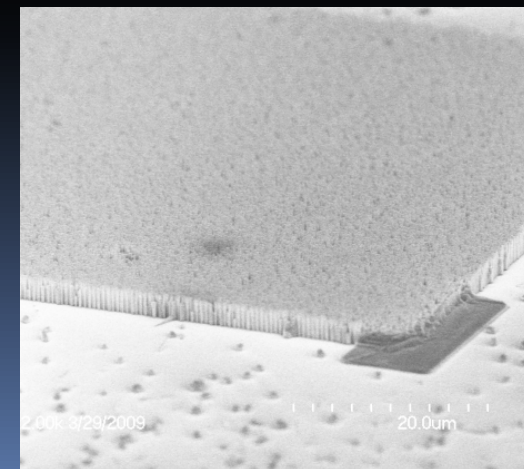
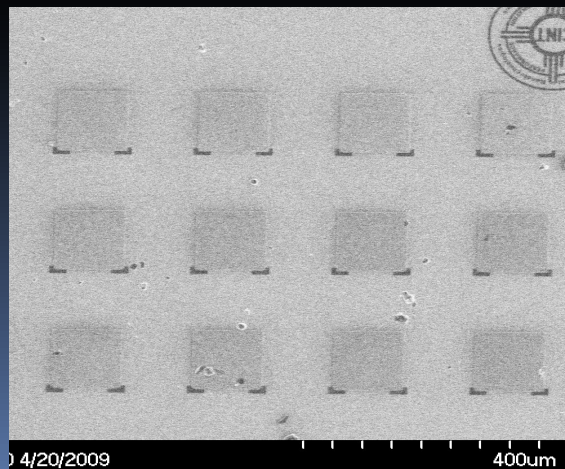
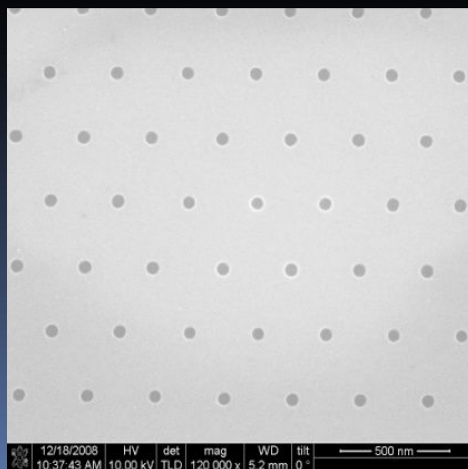


Device Fabrication

- *n*-doped GaAs nanopillars
 - Catalyst-free via selective-area epitaxy using MOCVD
 - 700° C, V-III ratio of 10:1, 20 minute planar growth at 1 A/s
- Hexagonal shaped, using a SiO₂ mask for patterning
- Grown in different arrays of constant height and width
 - Height range: 265 – 626 nm
 - Width range: 27-82 nm
- Pillars studied in an array with height 306 nm, radius 82 nm



10.0kV x50.0k 5/1

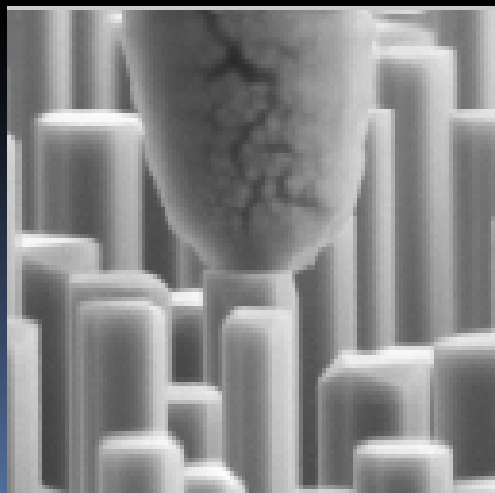


Doping & Passivation

- Device doped using Si for *n*-type
- Concentration determined using Hall measurements
 - Used different concentrations for calibration
 - Both passivated and unpassivated pillars used
- Doping level determined to be $7 \times 10^{18} \text{ cm}^{-3}$
- Passivation done using ammonium sulfide solution $(\text{NH}_4)_2\text{S}$, at 22% concentration
- Process created one monolayer of sulfur on As-terminated surface
- Same sample measured, passivated, and re-measured twice

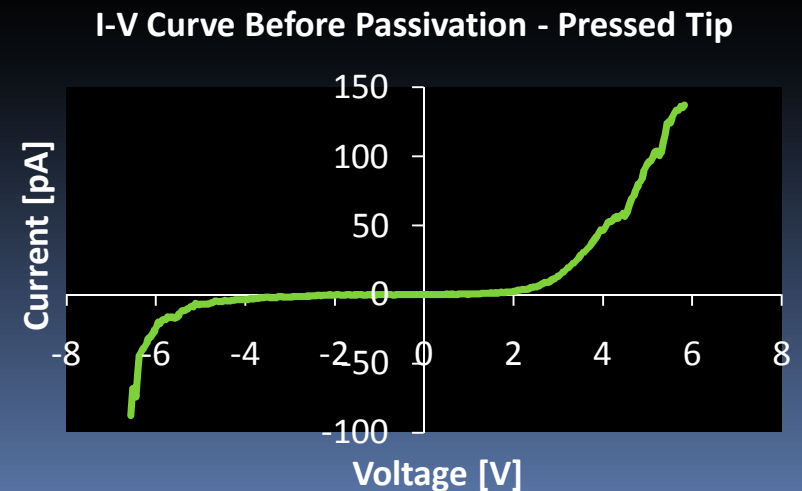
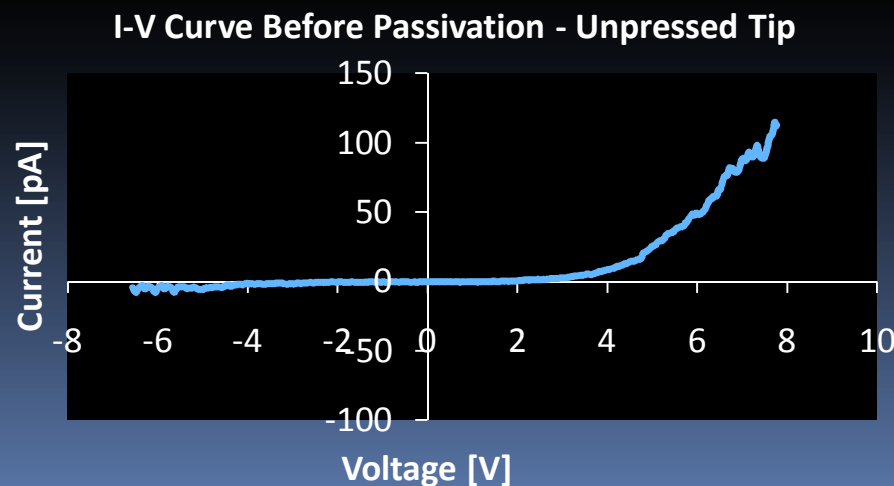
Measurement

- Pillars measured using AFM
 - Contact-mode, Au/Pt-coated tip
 - Pillars still on growth substrate
- Current measured with tip voltage of -10 to +10 V
- Sample placed on metal disc with silver epoxy, top of pillar probed with tip



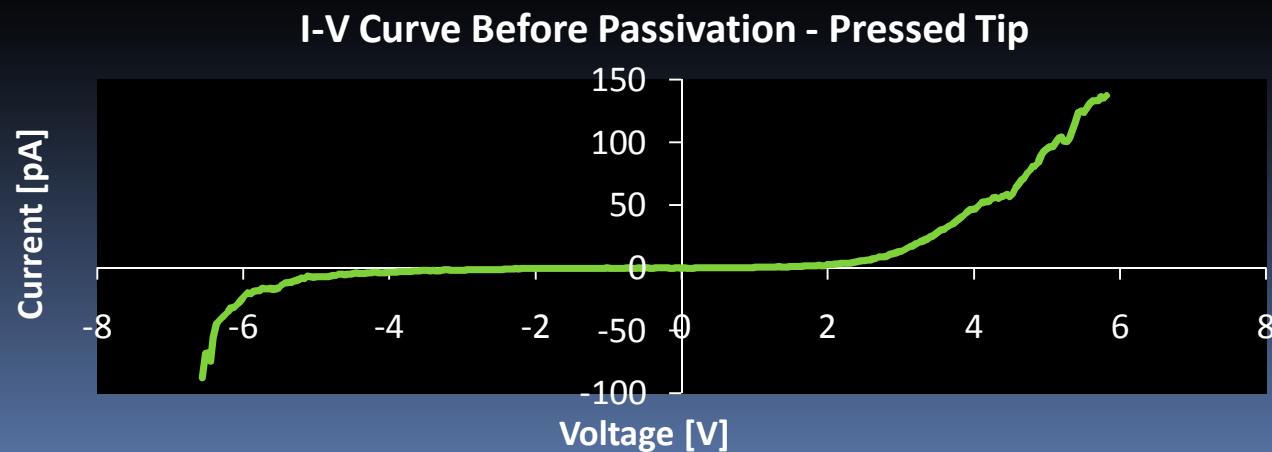
Results – No Passivation

- Tip simply contacting pillar top creates rectifying I-V Curve (diode-like)
 - Injection-limited current
 - Exponential I-V curve only at forward bias
- Pressing tip into pillar created a $I \propto V^2$ curve
 - Space-charge limited current, lower contact resistance
 - Possible destruction of native oxide, or curved tip creating a field enhancement



Results – No Passivation

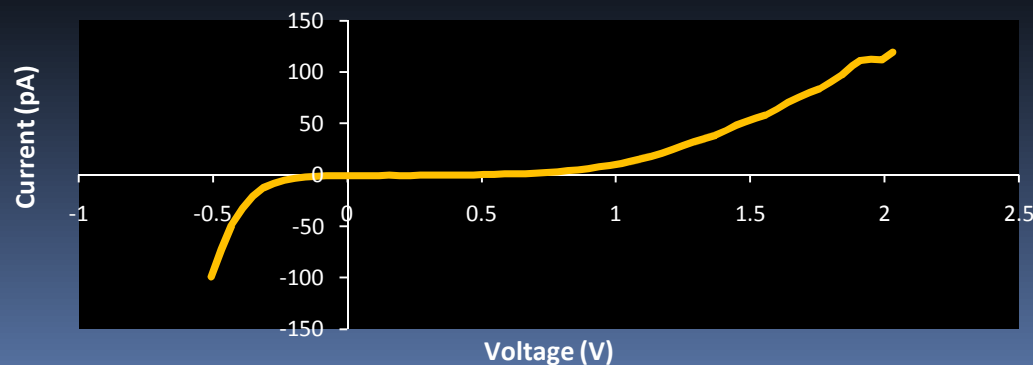
- Pressing tip in, average I-V measurements over 10 samples showed resistance of 44.9 G Ω
 - Using linear approximation of forward bias region
- Current flow begins at 2 V, reaches 50 pA at 4 V
- Maximum specific resistivity of $8 \times 10^5 \Omega\text{-cm}$
 - Average value of $3.9 \times 10^5 \Omega\text{-cm}$
 - Schottky barrier height 1.159 eV
- Pillar's transport severely limited by both surface state density and Schottky barrier



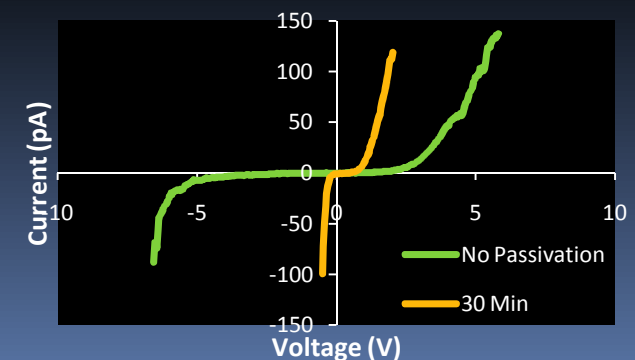
Results – 30 Min Passivation

- Pillars were passivated in $(\text{NH}_4)_2\text{S}$ for 30 min
- Similar $I \propto V^2$ curve obtained, but current flows at a lower bias
 - Current flow begins at 0.75 V, reaches 50 pA at 1.5 V
- Contact resistance still exists
 - Still due to Schottky barrier
- Average resistance 17 G Ω
- Average resistivity $1.17 \times 10^5 \Omega\text{-cm}$
- Further passivation needed to obtain a linear I-V curve

I-V Curve after 30 m Passivation

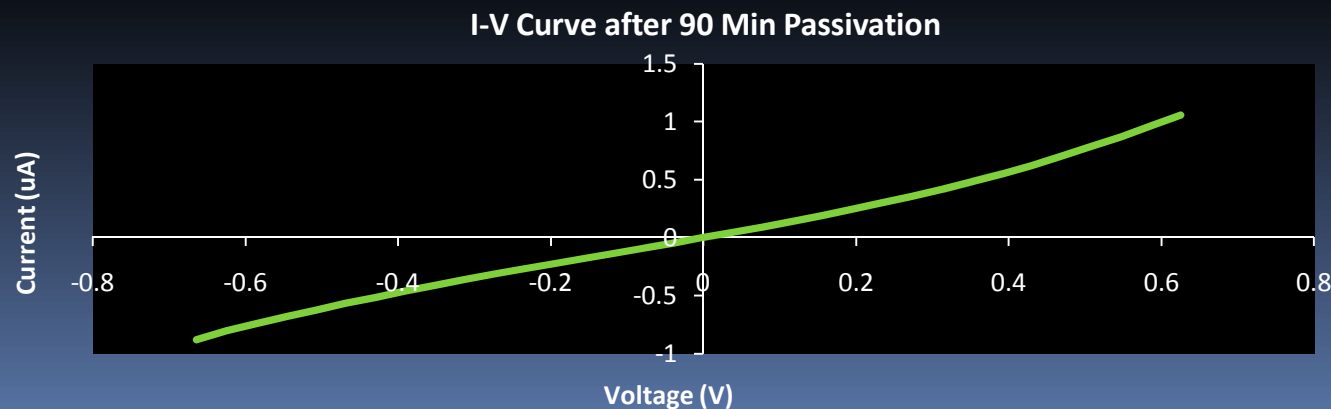


I-V Curve 0 and 30 m Passivation



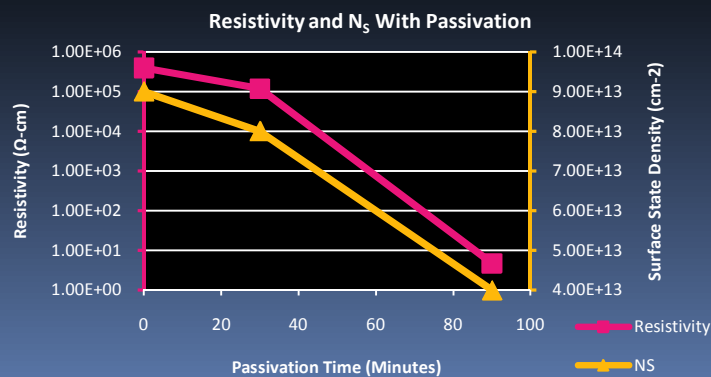
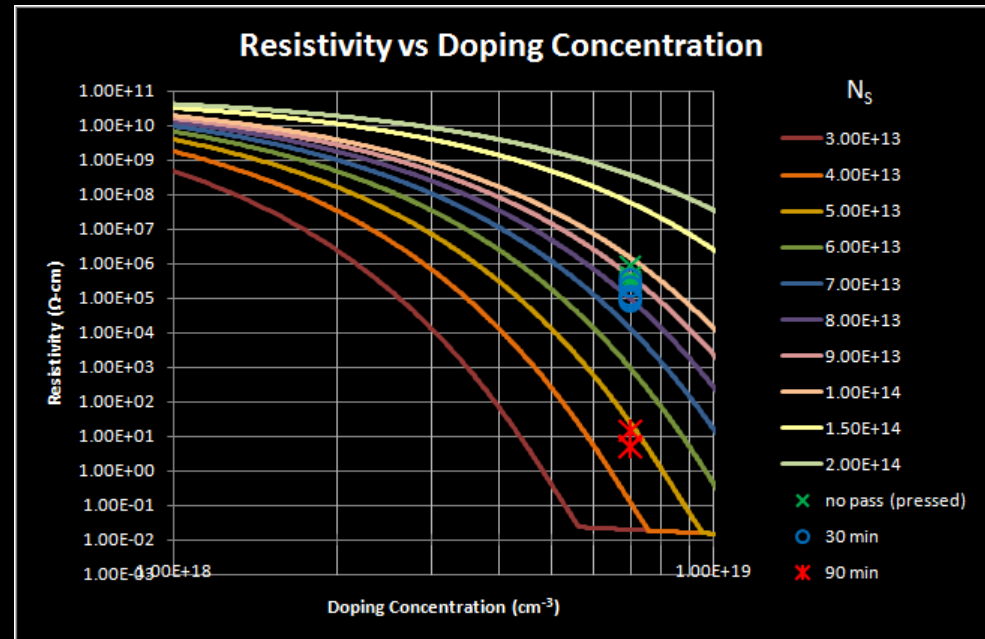
Results – 90 Min Passivation

- After 90 min passivation, I-V curve becomes linear
- Much more current flows at a much lower bias
 - Current is $1.06 \mu\text{A}$ at 0.625 V
 - Much higher current than the 30-min and non-passivated pillars
 - Suggests transport is no longer hindered by depletion region
 - Schottky barrier height only 0.546 eV
- Average resistance $724 \text{ k}\Omega$
- Average resistivity $4.7 \Omega\text{-cm}$



Overall Results – Surface State Density

- Surface state density is lowered with passivation
- 0 and 30 minute passivation densities overlap - slight decrease
- 90 minute passivation lowers by half an order of magnitude – large decrease
- High doping of $7 \times 10^{18} \text{ cm}^{-3}$ means small changes in N_s will be significant
- Resistivity approaches lowest possible calculated value of $10^{-2} \Omega\text{-cm}$



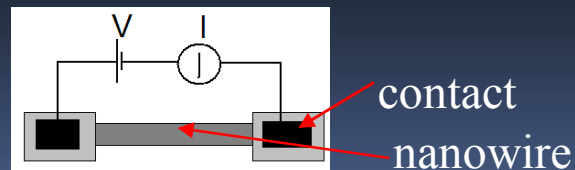
Passivation Time (min)	Resistance (Ω)	Resistivity ($\Omega\text{-cm}$)	Surface State Density (cm^{-2})
0	44.9×10^9	3.9×10^5	9×10^{13} – 1×10^{14}
30	17×10^9	1.17×10^5	8×10^{13}
90	724×10^3	4.7	4×10^{13}

Summary

- Devised a system of equations to relate resistivity to surface state density for n -doped GaAs nanopillars
 - Relationship with doping concentration, electronic radius, effective carrier concentration, resistivity
- Distinguished between physical radius and electronic radius
 - Depletion region created by surface states
- Used MATLAB to simulate these equations for nanopillars
 - Tied actual I-V data and resistivity from real nanopillars to model to determine their surface state density
- Observed and recorded the effects of passivation on nanopillars and their effect on surface states
 - Determined amount of surface state reduction
- A simple, fast and inexpensive method to determine the hard-to-measure parameter of surface state density

Future Work

- Further verify accuracy using a single-wire device
 - Single nanopillar off-substrate placed on metal contacts for Ohmic contact
 - Remove any possible effects from Schottky barrier created by AFM tip
- Fabrication and testing of device, passivated and non-passivated, is still underway



Works Cited

- [1] G. S. Chang, W. C. Hwang, Y. C. Wang, Z. P. Yang, and J. S. Hwang., "Determination of surface state density for GaAs and InAlAs by room temperature photoreflectance." *Journ. of Appl. Phys.*, 1999, Issue 3, Vol. 86.
- [2] V. Schmidt, S. Senz, and U. Gosele., "Influence of the Si/SiO₂ interface on the charge carrier density of Si nanowires." *Applied Physics A*, 2007, Vol. 86.
- [3] Bube, Richard H., "Electronic Properties of Crystalline Solids." New York : Academic Press, 1974.
- [4] K. Seo, S. Sharma, A. Yasseri, D. R. Stewart, and T. I. Kamins., "Surface charge density of unpassivated and passivated metal-catalyzed silicon nanowires." *Electrochemical and Solid-State Letters*, 2006, Issue 3, Vol. 9.
- [5] D. Stichtenoth, K. Wegener, C. Gutsche, I. Regolin, F. J. Tegude, W. Prose, M. Seibt, and C. Ronning., "P-type doping of GaAs nanowires." *Applied Physics Letters*, 2008, Issue 163107, Vol. 92.
- [6] F. Varurette, J. P. Nys, D. Deresmes, B. Grandidier, and D. Stievenard., "Resistivity and surface states density of n- and p-type silicon nanowires." *Journal of Vacuum Science Technology B*, 2008, Issue 3, Vol. 26.
- [7] M. T. Bjork, H. Schmid, J. Knoch, H. Riel and W. Riess., "Donor deactivation in silicon nanostructures." *Nature Nanotechnology*, 2009, Vol. 4.
- [8] Banerjee, Sanjay and Streetman, Ben., "Solid State Electronic Devices." Upper Saddle River, NJ : Pearson Education, Inc, 2006.
- [9] E. Yablonovitch, C. J. Sandroff, R. Bhat, and T. Gmitter., "Nearly ideal electronic properties of sulfide coated GaAs surfaces." *Appl. Phys. Lett.*, 1987, Issue 6, Vol. 51. 439-441.
- [10] Peng Jin, S.H. Pan, Y.G. Li, C.Z. Zhang, Z.G. Wang., "Electronic properties of sulfur passivated undoped-n+ type GaAs surface studied by photoreflectance." *Applied Surface Science*, 2003, Issue 218.
- [11] Sze, S. M., "Physics of Semiconductor Devices." New York : Wiley, 1981.
- [12] C. J. Sandroff, M. S. Hegde, L. A. Farrow, C. C. Chang, and J. P. Harbison., "Electronic passivation of GaAs surfaces through the formation of arsenic-sulfur bonds." *Applied Physics Letters*, 1989, Issue 4, Vol. 54, pp. 362-364.
- [13] A. Alec Talin Francois Leonard, B. S. Swartzentruber, Xin Wang, and Stephen D. Hersee., "Unusually Strong Space-Charge-Limited Current in Thin Wires." *Physical Review Letters*, 2008, Issue 7, Vol. 101.